

Dimensional effects on the tunneling conductivity of gold-implanted nanocomposite films

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We study the dependence of the electrical conductivity on the gold concentration of Au-implanted polymethylmethacrylate (PMMA) and alumina nanocomposite thin films. For Au contents larger than a critical concentration, the conductivity of Au-PMMA and Au-alumina is well described by percolation in two dimensions, indicating that the critical correlation length for percolation is larger than the thickness of the films. Below the critical loading, the conductivity is dominated by tunneling processes between isolated Au particles dispersed in PMMA or alumina continuous matrices. Using an effective medium analysis of the tunneling conductivity, we show that Au-PMMA behaves as a tunneling system in two dimensions, as the film thickness is comparable to the mean Au particle size. On the contrary, the conductivity of Au-alumina films is best described by tunneling in three dimensions, although the film thickness is only a few times larger than the particle size. We interpret the enhancement of the effective dimensionality of Au-alumina films in the tunneling regime as due to the larger film thickness as compared to the mean interparticle distances.

I. INTRODUCTION

Nanocomposite films made of nano-sized metallic particles embedded in an insulating matrix display peculiar electron transport regimes depending on the relative amount of the metallic phase compared to the insulating one. Above a critical concentration x_c of the metallic phase, macroscopic clusters of connected metallic particles span the entire sample, giving rise to an electrical conductivity σ characterized in the vicinity of x_c by a power-law behavior of the form:

$$\sigma \propto (x - x_c)^t, \quad (1)$$

where t is the transport exponent which the percolation theory predicts to take the values $t \simeq 1.3$ and $t \simeq 2$ for strictly two-dimensional (2d) and three-dimensional (3d) systems, respectively.^{1,2} Equation (1) describes well the conductivity behavior in the $x > x_c$ region of several nanogranular metal films grown by different methods,³⁻⁶ with observed values of the exponent that are consistent with the effective dimensionality of the films. Below x_c the metallic phase is broken up into disconnected metallic regions, so that for concentrations sufficiently smaller than x_c electrons have to tunnel across the insulating barrier separating the conducting fillers. In this regime, the conductivity of nanocomposites with embedded, isolated metallic particles follows approximately:⁷⁻¹⁰

$$\sigma \propto e^{-\frac{2\delta(x)}{\xi}}, \quad (2)$$

where ξ is the tunneling decay length which, depending on the nature of the composite constituents, ranges from a fraction to a few nanometers, and $\delta(x)$ is a typical distance between the surfaces of two conducting particles. In the dilute limit $x \rightarrow 0$, dimensional considerations¹¹ imply that $\delta(x) \propto D/x^{1/d}$ for particles of mean size D randomly dispersed in a d -dimensional volume, as

also inferred from transport measurements on 2d and 3d nanogranular metal composites.^{6,12}

Recently, a survey of the conductivity data of several thick nanogranular composite films has evidenced that the percolation behavior of Eq. (1) evolves into the tunneling one of Eq. (2) as the result of the competition between percolation and tunneling transport mechanisms as x decreases.¹³

In this article we study the conductivity behavior as a function of gold content in Au-implanted nanocomposite films above and below x_c . We consider two distinct series of systems: Au-implanted alumina and Au-implanted polymethylmethacrylate (PMMA) films. Au-alumina films have thicknesses h a few times larger than the mean Au particle size D , while the conducting layer of Au-PMMA is basically formed by a single layer of gold particles ($h \approx D$), which makes this system strictly two dimensional. We show that for $x > x_c$ the conductivity of both systems follow Eq. (1) with transport exponents very close to the 2d value $t \simeq 1.3$, in accord with the prediction of the percolation theory of nanocomposites with nanometric values of h . To study the tunneling behavior at Au concentrations below x_c , we analyze the conductivities of Au-PMMA and Au-alumina using an effective medium theory formulated for film thicknesses ranging from $h = D$ to $h \gg D$, as to describe the evolution from 2d to 3d of the conductivity. We find that, in contrast to the 2d nature of the Au-PMMA films, the Au-alumina films are large enough to sustain tunneling conductivity with 3d character. The effective dimensionality of our ion-implanted Au-alumina nanocomposites thus increases from 2d to 3d as the system crosses over from the percolation to the tunneling regimes. This conclusion is supported by the value of ξ extracted from the conductivity data, which coincides with that observed¹³ in thick (i.e., 3d) co-sputtered Au-alumina nanocomposite films.¹⁴

II. MATERIAL AND METHODS

The experimental data used in the present work are basically the same ones presented in Refs. 4, 15, and 16, but with a completely different approach and exploring a different range of the data. The mentioned works addressed specifically implantation doses above the percolation thresholds, while the present work focuses in doses below the percolation thresholds, where the particles are isolated from each other and the tunneling effect is dominant.

A summary of the experimental procedure will be done here and details can be obtained in the mentioned references.

PMMA was deposited on glass microscope slides using a spin coater, generating a film thickness about 50 nm. Electrical contacts were formed at both ends of the substrates by plasma deposition of thick (200 nm) platinum films. Very low energy (49 eV) was used for ion implantation using a streaming (unidirectionally drifting) charge-neutral plasma formed by a vacuum arc plasma gun.

Gold ion implantation in alumina using 40 keV was proceeded in an implanter.¹⁷ Electrical contacts were formed at both ends of the alumina sample using silver paint.

For both Au-PMMA and Au-alumina systems, in situ resistance measurements were performed as the ion implantation proceeded:^{4,15} after a known dose of Au ions is implanted, the implantation process is temporary halted and the resistance is measured. This process is repeated to determine the sample conductivity as a function of ion implantation dose.

In the present work, computer simulation using the TRIDYN computer code^{18,19} was used to estimate the depth profiles of the ion implanted gold in the alumina substrate. TRIDYN is a Monte Carlo simulation program based on the TRIM (Transport and Range of Ions in Matter) code.²⁰ The program takes into account compositional changes in the substrate due to: previously implanted dopant atoms, and sputtering of the substrate surface. Note that, the parameters t (transport exponent) and x_c (critical concentration of the metallic phase) were recalculated in the present work, using a different approach, presenting some deviation from those obtained in the previous works.^{4,15}

III. RESULTS

Results of our conductivity measurements on Au-PMMA and Au-alumina nanocomposites are shown in Fig. 1 (open squares), where the conductivity ratio σ/σ_0 is plotted as a function of the normalized Au concentration $x = \varphi/\varphi_0$. The saturation dose φ_0 is defined as the implantation dose φ above which a continuous metal film starts to be deposited on the insulator surface, while the saturation conductivity σ_0 is the conductivity mea-

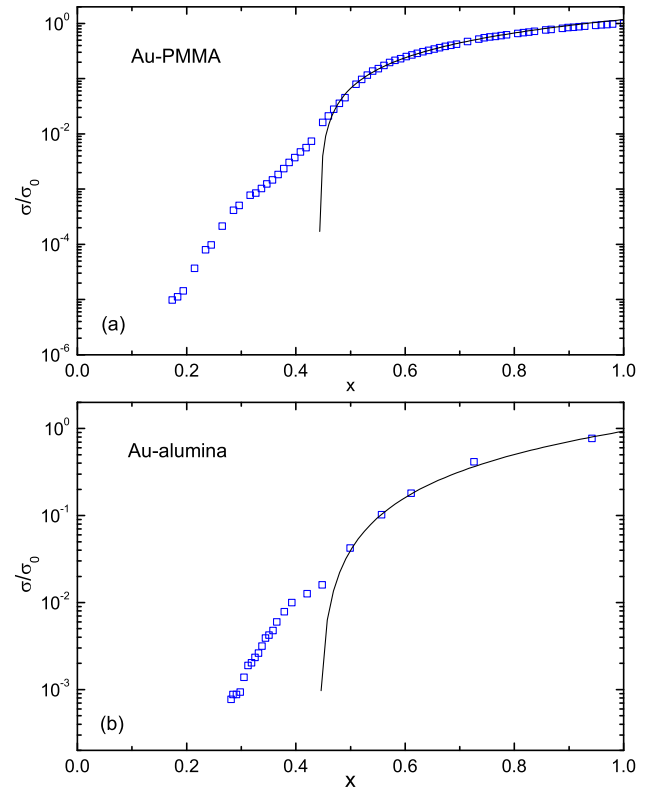


FIG. 1. (Color online) Normalized conductivity (open squares) of (a) Au-PMMA and (b) Au-alumina nanocomposite films as a function of the normalized ion dose x . Solid lines are fits with Eq. (1).

sured at $\varphi = \varphi_0$ (i.e., at $x = 1$). We have determined $\varphi_0 = 2 \times 10^{16}$ atoms cm^{-2} and $\sigma_0 = 2 \times 10^6$ S/m for Au-PMMA,⁴ and $\varphi_0 = 9.5 \times 10^{16}$ atoms cm^{-2} and $\sigma_0 = 14$ S/m for Au-alumina.¹⁵ As is apparent from the semi-log plot of Fig. 1, the x -dependence of σ/σ_0 for both Au-PMMA and Au-alumina films presents a double hump shape, which is a feature commonly observed also in several other nanogranular metals composites.¹³ The hump at values of x larger than 0.4–0.5 stems from the presence of clusters of coalesced Au particles that extend across the entire composite layer. In this region, σ/σ_0 is reasonably well fitted by Eq. (1) (shown by solid lines in Fig. 1) with $t = 1.26 \pm 0.03$ and $x_c = 0.443 \pm 0.002$ for Au-PMMA, and $t = 1.4 \pm 0.2$ and $x_c = 0.44 \pm 0.02$ for Au-alumina composite films.

The fitted values of the transport exponents are both consistent with the 2d value $t \simeq 1.3$, indicating that ion-implanted Au-PMMA and Au-alumina composites behave as 2d percolating composites for $x > x_c$. This result is consistent with the values of the thickness of the conducting layers of Au-PMMA and Au-alumina extracted from the TRYDIN analysis and from cross sectional TEM images of Au-PMMA. For the case of Au-PMMA films we observed a conducting layer of thickness $h \approx 5.5 - 8$ nm (see Fig. 2(a)) formed by Au particles of size $D \approx 5 - 6$

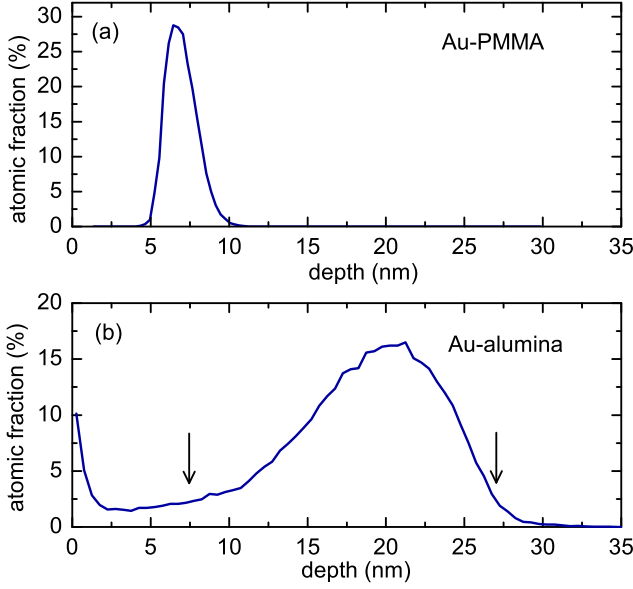


FIG. 2. (Color online) TRIDYN simulations of the depth profiles for gold implanted into (a) PMMA with ion energy 49 eV and dose $\varphi = 0.4 \times 10^{16}$ atoms cm^{-2} , and into (b) alumina with ion energy 40 keV and dose $\varphi = 2.5 \times 10^{16}$ atoms cm^{-2} . With these doses, the systems are below their respective percolation thresholds, presenting isolated gold nanoparticles. The arrows in (b) delimit 20 nm that is the conducting layer thickness considered.

nm,¹⁶ which indicates that Au-PMMA films are composed basically of one monolayer of Au particles embedded into the PMMA substrate. Au-implanted PMMA films are thus strictly 2d systems. On the contrary, the conducting layer of Au-alumina films is not strictly two-dimensional: the mean Au particle size extracted from TEM is $D \approx 3.2$ nm,¹⁵ while TRYDIN analyses indicate that $h \approx 20$ nm (see Fig. 2(b)), that is, about 6 times larger than D . The 2d character of the percolation conductivity of Au-alumina films is explained by observing that the relevant length scale that governs the power-law behavior of Eq. (1) is the correlation length ζ (not to be confused with the tunneling decay length ξ), which measures the typical size of finite clusters of connected particles.¹ When x approaches x_c from either above or below the percolation threshold x_c , the correlation length increases as $\zeta \approx a|x - x_c|^{-\nu}$, where a is of the order of the particle size D , and $\nu > 0$ is the correlation length exponent. For concentrations such that ζ becomes larger than the film thickness h , the composite behaves effectively as a 2d system,^{21,22} so that in the vicinity of x_c the conductivity is expected to follow Eq. (1) with $t \approx 1.3$, as observed in our Au-alumina samples.²³

Concerning the percolation thresholds extracted from the fits, we note that the values of x_c of Au-PMMA and Au-alumina are basically the same, although the percolation threshold is generally expected to depend on the particular combination of metal and insulator compo-

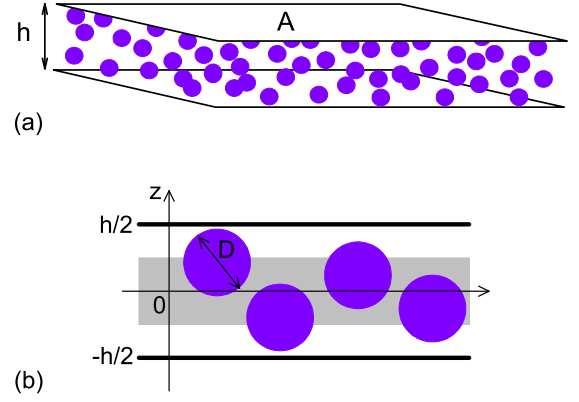


FIG. 3. (Color online) Model of a nanocomposite film with finite thickness of the conducting layer. (a) The metallic particles are taken to be spheres with identical diameter D dispersed within the space delimited by two parallel hard planes of area A and separated by a distance gh . The spheres cannot penetrate the confining walls. (b) Cross-sectional view illustrating the volume $V_h = (h - D)A$ available for the sphere centers (grey region).

nents constituting the films and on the details of the fabrication process.^{13,14} We have no arguments ruling out that the equivalence of the two percolation thresholds is just a mere coincidence, as very similar thresholds have already been observed in granular films with different components.¹³ Interestingly, however, the value $x_c \approx 0.44$ of our Au-alumina films is larger than the percolation threshold of three-dimensional Au-alumina composites ($x_c \approx 0.38$), which is consistent with the observation that the percolation threshold generally diminishes as the dimensionality increases.²

For values of x lower than x_c clusters of coalesced particles no longer span the entire composite layer and the conductivity is dominated by tunneling processes, which give rise to the hump of σ at $x < x_c$ in Fig. 1. In this region the conductivities of Au-PMMA and of Au-alumina films are expected to follow Eq. (2) with typical inter-particle distances $\delta(x)$ whose x -dependence is influenced by the effective dimensionality of the system. Although for the case of Au-PMMA films, which are strictly two-dimensional, $\delta(x)$ can be determined by considering dispersions of conducting particles in a plane, for Au-alumina films one must consider the effect of the finite width h of the conducting layer, as its observed value ($h \approx 6D$) is such that tunneling processes between Au particles at different depths within the conducting layer are also possible.

To tackle this problem, we consider an effective medium approximation (EMA) for the tunneling conductivity applied to a dispersion of N spherical particles of diameter D that are confined by two parallel hard walls separated by a distance $h \geq D$ and of macroscopic area A , as shown schematically in Fig. 3(a). The EMA con-

ductivity \bar{g} is the solution of the following equation:^{10,13}

$$\frac{1}{N} \left\langle \sum_{i \neq j} \frac{g(r_{ij})}{\bar{g} + g(r_{ij})} \right\rangle = 2, \quad (3)$$

where $g(r_{ij}) = g_0 \exp[-2(r_{ij} - D)/\xi]$, with $r_{ij} \geq D$, is the tunneling conductance between two spherical particles i and j , $r_{ij} = |\vec{r}_i - \vec{r}_j|$ is the distance between their centers located at \vec{r}_i and \vec{r}_j , and g_0 is a tunneling prefactor. The angular brackets in Eq. (3) denote a statistical average over the positions of the spheres occupying the volume delimited by the two parallel planes located at $z = \pm h/2$, as shown in Fig. 3(b). Since the spherical particles cannot penetrate the hard walls, the available volume for the sphere centers is $V_h = (h - D)A$. Assuming that the particles are uncorrelated, the average reduces to:

$$\langle (\dots) \rangle = \frac{1}{V_h^N} \int_{V_h} d\vec{r}_1 \dots \int_{V_h} d\vec{r}_N (\dots), \quad (4)$$

so that Eq. (3) can be rewritten as:

$$\frac{N}{V_h^2} \int_{V_h} d\vec{r}_1 \int_{V_h} d\vec{r}_2 \frac{\theta(r - D)}{g^* \exp[2(r - D)/\xi] + 1} = 2, \quad (5)$$

where $g^* = \bar{g}/g_0$, $\theta(x) = 1$ for $x \geq 0$ and $\theta(x) = 0$ for $x < 0$, and $r = |\vec{r}_1 - \vec{r}_2|$. Using

$$\iint_{-(h-D)/2}^{(h-D)/2} dz_1 dz_2 = \int_{-\infty}^{+\infty} dz (h - D - |z|) \theta(h - D - |z|), \quad (6)$$

where $z = z_1 - z_2$, we express Eq. (5) in terms of an integral over $\vec{r} = \vec{r}_1 - \vec{r}_2$ which, due to the exponential decay of the tunneling conductance, can be extended over the whole space:

$$\frac{h\rho}{h - D} \int d\vec{r} \theta(r - D) \frac{(h - D - |z|) \theta(h - D - |z|)}{g^* \exp[2(r - D)/\xi] + 1} = 2, \quad (7)$$

where $\rho = N/(hA)$ is the particle number density. Finally, we pass to spherical coordinates and integrate over the angles to find that the EMA conductivity g^* satisfies:

$$\frac{12h\eta}{D^3} \int_D^\infty dr \frac{r}{g^* \exp[2(r - D)/\xi] + 1} = 2 \quad (8)$$

for $D \leq h \leq 2D$, and

$$\begin{aligned} & \frac{12h\eta}{D^3} \int_{h-D}^\infty dr \frac{r}{g^* \exp[2(r - D)/\xi] + 1} \\ & + \frac{24h\eta}{(h - D)D^3} \int_D^{h-D} dr \frac{r^2}{g^* \exp[2(r - D)/\xi] + 1} \\ & - \frac{12h\eta}{(h - D)^2 D^3} \int_D^{h-D} dr \frac{r^3}{g^* \exp[2(r - D)/\xi] + 1} = 2 \end{aligned} \quad (9)$$

for $h \geq 2D$. In Eqs. (8) and (9) we have introduced the dimensionless density $\eta = \pi D^3 \rho / 6$ which, for the

case of uncorrelated (i.e., penetrable) spheres, is related to the volume fraction x of the metallic phase through $x = 1 - \exp(-\eta)$.²⁴ The 2d and 3d limits are obtained by setting, respectively, $h = D$ in Eq. (8) and $h \gg D$ in Eq. (9). It is instructive to express g^* as a tunneling conductance between the surfaces of two spheres separated by a characteristic distance δ^* :

$$g^* = e^{-\frac{2\delta^*}{\xi}}. \quad (10)$$

In this way, the term $1/\{g^* \exp[2(r - D)/\xi] + 1\}$ that appears in the integrands of Eqs. (8) and (9) reduces for $\xi/D \ll 1$ to:

$$\frac{1}{g^* \exp[2(r - D)/\xi] + 1} = \frac{1}{\exp[2(r - D - \delta^*)/\xi] + 1} \approx \theta(\delta^* + D - r). \quad (11)$$

Using Eq. (11) in Eq. (8), we find thus that the characteristic distance δ^* for a strictly 2d system becomes:

$$\frac{\delta^*}{D} = \left[\frac{1}{3 \ln(1 - x)^{-1}} + 1 \right]^{1/2} - 1, \quad (12)$$

while, since the first and third integrals in Eq. (9) vanish for $h \gg D$, for a 3d system we obtain:

$$\frac{\delta^*}{D} = \left[\frac{1}{4 \ln(1 - x)^{-1}} + 1 \right]^{1/3} - 1. \quad (13)$$

For $x \ll 1$ the above expressions correctly reproduce the dimensional scaling $\delta^* \propto D/x^{1/d}$ expected to hold true in the dilute limit. Furthermore, Eqs. (12) and (13) evidence that, for moderately small values of x , the relevant tunneling distance for 2d systems is about twice that for 3d systems, as illustrated in Fig. 4(a) where Eqs. (12) and (13) are shown by solid lines.

To assess how the EMA conductivity evolves from the 2d to the 3d limits, we solve numerically Eqs. (8) and (9) for films thicknesses ranging from $h = D$ to $h \gg D$. Expressing the resulting g^* in terms of the distance δ^* as specified in Eq. (10), we find that δ^* rapidly decreases as the film thickness increases from $h = D$, and essentially matches the 3d limit already for $h \gtrsim 3D$, as shown in Fig. 4(b) where δ^* is presented as a function of h/D for selected x values. The rapid evolution from 2d to 3d is also illustrated in Fig. 4(a), where the numerical values of δ^* as function of x and for several film thicknesses are compared with Eqs. (12) and (13).

The above analysis suggests that it suffices to have film thicknesses only a few times larger than the particle size to induce a 3d character to the tunneling conductivity. This result is consistent with the observation that the relevant length scale for tunneling is given by the mean inter-particle distance, and has interesting consequences with respect to the nanocomposite films considered here. Indeed, if on the one hand Au-PMMA films are expected to display 2d conductivity in the whole range of x because $h \approx D$, on the other hand the conductivity of Al-alumina nanocomposites should be understood as having

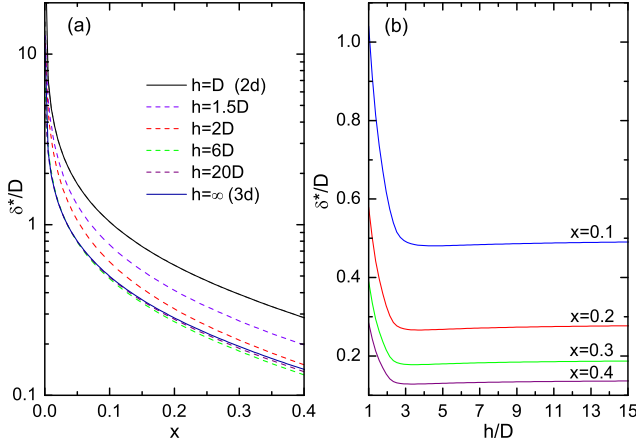


FIG. 4. (Color online) (a) Relevant tunneling distance δ^* between the surfaces of two spheres as a function of the metallic volume fraction x and for different thicknesses h of the film. δ^* is calculated from $\delta^* = (\xi/2) \ln(1/g^*)$, where g^* is the numerical solution of the EMA equations (8) and (9) with tunneling decay length fixed at $\xi = 0.05D$. Solid lines are results for the limits 2d and 3d given in Eqs. (12) and (13). (b) δ^* as a function of h for selected values of the volume fraction of the metallic spheres. For $\delta^* \gtrsim 3D$ the films behave practically as 3d systems.

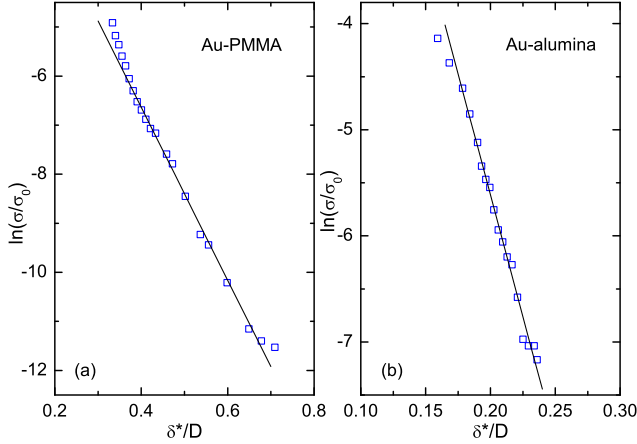


FIG. 5. (Color online) Natural logarithm of measured σ/σ_0 (open squares) as a function of δ^*/D for (a) Au-PMMA and (b) Au-alumina nanocomposite films. The characteristic EMA distance δ^* is assumed to be given by the 2d limit of Eq. (12) for Au-PMMA and by the 3d limit of Eq. (13) for Au-alumina. Solid lines are the best fits of the data with Eq. (14).

2d character for $x > x_c$ but 3d character for $x < x_c$. This is so because the measured thickness of the conducting layer of Au-alumina ($h \approx 6D$) is sufficiently larger than the EMA 2d-3d crossover value $h \approx 3D$ (see Fig. 4) to induce 3d tunneling.

Assuming that the above considerations capture the essential physics of the problem, we interpret the mea-

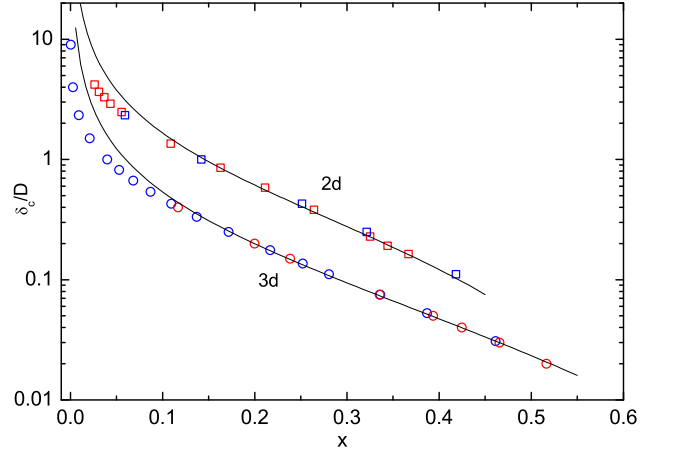


FIG. 6. (Color online) Monte Carlo results (open symbols) for the critical distance δ_c calculated for dispersions of impenetrable disks in 2d,^{27,28} and impenetrable spheres in 3d.^{9,26} Solid lines are $\delta_c = 2.5(r_{NN}^{2d} - D)$ for the 2d case and $\delta_c = 1.65(r_{NN}^{3d} - D)$ for the 3d case, where r_{NN}^{2d} and r_{NN}^{3d} are the expressions of the corresponding mean distances between the centers of nearest neighboring particles given in Eqs. (15) and (16).

sured conductivity data of Au-PMMA and Au-alumina films in terms of Eq. (2), where $\delta(x)$ is identified with δ^* as given by Eqs. (12) and (13), respectively. Consistently with the exponential form of Eq. (2), the $\ln(\sigma/\sigma_0)$ data of both systems follow approximately a linear dependence as a function of δ^* , as shown in Fig. 5. From linear fits of $\ln(\sigma/\sigma_0)$ with

$$\ln(\sigma/\sigma_0) = -\frac{2D}{\xi} \frac{\delta}{D} + \text{const.}, \quad (14)$$

we extract the values of $2D/\xi$ that best fit the data to find $\xi/D = 0.113 \pm 0.002$ for Au-PMMA and $\xi/D = 0.044 \pm 0.001$ for Au-alumina. Remarkably, the value of ξ/D found for Au-alumina coincides with the value extracted from analyses of co-sputtered Au-alumina granular thick films,¹⁴ which are three-dimensional systems. Using the estimated mean size of Au particles in our films ($D \approx 3.2$ nm) we find $\xi \approx 0.14$ nm, which compares fairly well with $\xi \approx 0.1$ nm, obtained from $\xi = \hbar/\sqrt{2m\Delta E}$, where m is the electron mass and $\Delta E \approx 4$ eV is the estimated barrier height for tunneling between Au and alumina.¹³ It is worth noting that fitting the Au-alumina conductivity data using Eq. (14) with δ^* as given by the 2d limit of Eq. (12) gives $\xi \approx 0.3$ nm, which is about twice the value found assuming 3d tunneling. From $\xi/D \approx 0.113$ found for Au-PMMA films and from the corresponding mean Au particle size ($D \approx 5 - 6$ nm) we find $\xi \approx 0.6 - 0.7$ nm, which indicates that the tunneling decay length of Au-PMMA is substantially larger than that of Au-alumina nanocomposites.

To assess the robustness of the results based on the EMA approach, we repeat the above analysis by identi-

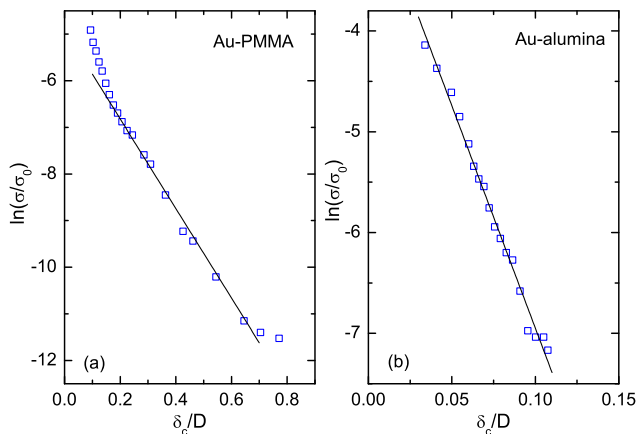


FIG. 7. (Color online) Natural logarithm of measured σ/σ_0 (open squares) as a function of δ_c for (a) Au-PMMA and (b) Au-alumina nanocomposite films. The critical distance δ_c is assumed to be given by the 2d limit for Au-PMMA and by the 3d limit for Au-alumina. Solid lines are the best fits of the data with Eq. (14).

fying δ of Eq. (2) with the critical distance δ_c , as prescribed by the critical path approximation (CPA).^{7-9,25} According to CPA, δ_c is defined as the shortest among the inter-particle distances $\delta_{ij} = r_{ij} - D$ such that the set of bonds satisfying $\delta_{ij} \leq \delta_c$ forms a percolating cluster. For dispersions of metallic particles in which the distances δ_{ij} span several multiples of ξ , CPA ensures that Eq. (2) with $\delta = \delta_c$ gives a good estimate of the composite conductivity.⁹

Monte Carlo results of δ_c for impenetrable spheres dispersed in a 3d volume^{9,26} and impenetrable disks dispersed in a 2d area^{27,28} are shown in Fig. 6 by open symbols. For the 2d case, we set $x = 2\phi_{2d}/3$ to convert the area fraction ϕ_{2d} covered by the disks to the volume fraction x of spheres of equal diameter with centers lying on a plane. In analogy with the EMA results of Fig. 4, the critical distance for the 2d case is systematically larger than the critical distance in 3d. For x larger than about 0.1 the Monte Carlo data are well reproduced by setting $\delta_c = 2.5(r_{NN}^{2d} - D)$ for the 3d case and $\delta_c = 1.65(r_{NN}^{3d} - D)$ for the 3d case (solid lines in Fig. 6), where

$$r_{NN}^{2d} = D + D \frac{(1 - 3x/2)^2}{6x(2 - 3x/2)}, \quad (15)$$

and

$$r_{NN}^{3d} = D + D \frac{(1 - x)^3}{12x(2 - x)}, \quad (16)$$

are the mean distances between the centers of nearest neighboring spheres in 2d and 3d, respectively.^{24,29} We fit the $\ln(\sigma/\sigma_0)$ data of Au-PMMA and Au-alumina with Eq. (14) using respectively the 2d and 3d functional dependences of δ_c , as shown in Fig. 7. From the slopes of the straight lines we extract $\xi/D = 0.208 \pm 0.004$ for Au-PMMA and $\xi/D = 0.045 \pm 0.001$ for Au-alumina. We

immediately see that for the Au-alumina films the estimates of ξ/D from CPA and EMA coincide within errors, while those for Au-PMMA films differ by almost a factor 2. This discrepancy could be attributed to possible effects of local particle correlations, neglected within our EMA approach but fully accounted for in CPA, which are expected to be more prominent in 2d than in 3d. Combining the estimates from EMA and CPA, and using the measured mean size of Au particles, we infer that in 2d Au-PMMA films the tunneling decay length is $\xi \approx 0.6 - 1.2$ nm.

IV. CONCLUSIONS

We have presented a study of the dependence of the electrical conductivity on the gold concentration in Au-implanted composite thin films with different insulating matrices. We have evidenced that the film thickness may influence substantially the behavior of the conductivity below the percolation threshold x_c , where tunneling between isolated gold particles dominates. Specifically, we have shown that an effective medium theory predicts a crossover from two-dimensional to three-dimensional tunneling behavior when the film thickness h is larger than only about 3 times the mean Au particle size D . Au-implanted PMMA films, that have thickness $h \approx D$, are thus strictly 2d systems in the tunneling regime, while Au-implanted alumina films are expected to show 3d tunneling behavior, as for this system $h \approx 6D$. Interestingly, the dimensionless tunneling decay length ξ/D extracted from the tunneling conductivity data of Au-implanted alumina films coincides with previous estimates of ξ/D in co-sputtered Au-alumina thick film composites. We have also shown that above the percolation threshold the measured conductivity of both Au-PMMA and Au-alumina follows a percolation power-law behavior with 2d transport exponent, in accord with the theory of percolation in thin disordered films. Au-PMMA films have thus 2d character in the whole range of Au concentrations, while the effective dimensionality of Au-implanted alumina films increases from 2d to 3d as the system crosses over from the percolation regime to the tunneling regime. These results and interpretations could find a firmer confirmation by measuring the conductivity behavior in films with thicknesses ranging continuously from $h \approx D$ to $h \gg D$. From our model we expect that the tunneling conductivity crosses over from two-dimensional to three-dimensional behaviors when the film thickness is about $2D$ - $3D$, as inferred from the behavior shown in Fig. 4.

ACKNOWLEDGMENTS

This work was supported by the Fundação de Amparo à Pesquisa do Estado de São Paulo (FAPESP) and the Conselho Nacional de Desenvolvimento Científico e Tecnológico (CNPq), Brazil. We are grateful to the In-

stitute of Ion Beam Physics and Materials Research at

the Forschungszentrum Dresden-Rossendorf, Germany, for the TRIDYN-FZR computer simulation code.

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